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Ts.I. Dimitrov^a, *R.H. Titorenkova*^b, *A.V. Zaichuk*^c, *Y.K. Tzvetanova*^b**SYNTHESIS AND STUDY OF FERRI-DIOPSIDE CERAMIC PIGMENTS**^a University of Ruse «Angel Kanchev», Branch Razgrad, Bulgaria^b Institute of Mineralogy and Crystallography «Acad. I. Kostov», Bulgarian Academy of Sciences, Sofia, Bulgaria^c Ukrainian State University of Chemical Technology, Dnipro, Ukraine

The article reports the mineralogical composition, structural features and changes in color parameters of iron-containing diopside pigments. The source of SiO₂ in these pigments was SiO₂·nH₂O with a particle size in the range of 2–7 μm, which is much more reactive than conventionally used quartz sand. A series of ceramic pigments on the basis of stoichiometric diopside in the system CaO–Fe₂O₃–MgO–SiO₂ was synthesized via solid-state high temperature sintering at 1000, 1100 and 1200°C. The resulting ceramic pigments were examined by powder X-ray diffraction, infrared spectroscopy, electron microscopy, electron paramagnetic resonance and Mössbauer spectroscopy. The color characteristics were measured using spectrophotometry method. It was found that multiphase ceramic pigments were obtained under the synthesis conditions, which contain diopside, wollastonite, cristobalite, akermanite and periclase in various proportions. The synthesized ceramic pigments are brown and beige in color, depending on the content of Fe₂O₃ additive, firing temperature and phase composition. The synthesized pigments can be used for staining of ceramic glazes.

Keywords: ceramic pigments, diopside, mineral composition, colorimetric indices, synthesis.**DOI:** 10.32434/0321-4095-2022-140-1-39-46**Introduction**

The ceramic pigments exhibit high corrosion resistance to the high-temperature action of glass melts of various natures, which makes it possible to achieve almost any color gamut of glass coatings [1].

The ceramic pigments are mainly obtained by the method of solid-phase synthesis from chemically pure reagents or natural raw materials [2,3]. Numerous studies have also been carried out [4–7] aimed at the effective use of various industrial wastes in the technology of ceramic pigments. At the same time, the synthesis of ceramic pigments, as a rule, is performed at high temperatures and requires the introduction of mineralizing additives [1].

The ceramic pigments of various silicate structures often require less energy consumption for the synthesis process due to the peculiarities of their structure, and also allow achieving a wide color palette [8–10].

The choice of a pyroxene matrix for ceramic pigments in this study is determined by that the

synthesis of diopside can be carried out at relatively low temperatures without using mineralizers. In this case, a stable color is formed, which depends on the nature of the introduced chromophore [11].

The color of the most natural and synthetic minerals is associated with the presence of transitional metals incorporated in the crystal structure, which have an unfilled d- or f-electron orbital determining the electronic transition under the action of light energy [1]. The crystal structure of diopside is monoclinic, space group C2/c. The pyroxene structure consists of single chains of tetrahedra extending parallel to c axis. Two types of cation positions are denoted as M1 and M2. The general formula of pyroxenes is M₂M₁T₂O₆. The following cations, Mg²⁺, Fe²⁺, Ca²⁺, Li⁺, and Na⁺, are in the M2 structural position in distorted polyhedron with coordination from 6 to 8. The following cations, Al³⁺, Fe³⁺, Fe²⁺, Ti³⁺, Cr³⁺, V³⁺, etc., are in the M1 octahedral position. Ions Si⁴⁺ and Al³⁺ are in tetrahedral position forming single chain. Natural diopside commonly contains various amounts of the

ions Fe^{2+} and forms complete solid solution with hedenbergite ($\text{CaFe}^{2+}\text{Si}_2\text{O}_6$) due to similar charge and ionic size of Mg^{2+} and Fe^{2+} [12].

The crystal structure of a synthetic $\text{CaFe}^{3+}\text{SiAlO}_6$ pyroxene was determined by single crystal X-ray diffraction [13]. One of the most interesting aspects of the synthetic $\text{CaFe}^{3+}\text{SiAlO}_6$ structure is the fact that Fe^{3+} ions are not concentrated solely at the octahedral M1 site, but are partitioned between the M1 site and the tetrahedral T site.

We suggest that the incorporation of iron into diopside structure in different structural positions can lead to synthesis of ceramic pigments.

Thus, the purpose of the study is to synthesize ceramic pigments based on stoichiometric diopside in the system $\text{CaO}-\text{Fe}_2\text{O}_3-\text{MgO}-\text{SiO}_2$ with various content of iron oxide additive and study the change in mineralogical composition, structural features and changes in color parameters of the resulting pigments.

Experimental

For the preparation of the diopside ceramic pigments in the system $\text{CaO}-\text{Fe}_2\text{O}_3-\text{MgO}-\text{SiO}_2$, the starting compositions were determined from the basic mineral diopside following the expression $\text{CaO}\cdot x\text{Fe}_2\text{O}_3\cdot(1-x)\text{MgO}\cdot 2\text{SiO}_2$, where $x=0.1, 0.2, 0.3, 0.4$ and 0.5 mole. Ceramic pigments were synthesized via solid-state high temperature sintering.

Starting materials used for the synthesis were CaCO_3 , Fe_2O_3 , MgO and $\text{SiO}_2\cdot n\text{H}_2\text{O}$ with particle size in the range of $2-7\ \mu\text{m}$, which is much more reactive than conventionally used quartz sand as a source of SiO_2 . Calculated quantities of materials for 100 g batch were weighed with a precision, then mixed and dry homogenized in a planetary mill Pulverizete-6 (Fritch). Synthesis was carried out in a laboratory muffle furnace in porcelain crucibles

with at the heating rate of $300-400^\circ\text{C}/\text{h}$ in air with isothermal retention during 2 hour at the final temperature. The resulting powder mixtures were sintered at 1000, 1100 and 1200°C in order to obtain Fe-doped diopside.

The resulting ceramic pigments were examined by powder X-ray diffraction (XRD) analysis, infrared spectroscopy (FT-IR), electron paramagnetic resonance (EPR), Mössbauer spectroscopy. The color was determined by spectrophotometry technique.

The phase composition of the synthesized ceramic pigments was determined using an X-ray diffractometer Empyrean, Malvern Panalytical operating at 40 kV and 30 mA with CuK_α radiation.

FT-IR spectra were collected using a Tensor 37 spectrometer (Bruker) with $4\ \text{cm}^{-1}$ resolution after averaging 128 scans on standard KBr pallets in the spectral region of $400-4000\ \text{cm}^{-1}$ at room temperature.

The Bruker EMXplus Spectrometer System ($9.4\ \text{GHz}$ – X-band region) was used to perform EPR analysis.

Mössbauer spectra were recorded by a Wissel (Wissenschaftliche Elektronik GmbH, Germany) electromechanical spectrometer working in a constant acceleration mode at room temperature (RT). A $^{57}\text{Co}/\text{Rh}$ source (activity of $\approx 20\ \text{mCi}$) and an $\alpha\text{-Fe}$ standard were used.

The colors of the pigments were determined by Lovibont Tintometer RT 100 Color and presented in the CIELab color space.

Results and discussion

X-ray diffraction (XRD) analysis

The powder XRD data revealed that the ceramic pigments synthesized at different temperature and Fe_2O_3 content vary in their phase composition

Table 1

Results of powder XRD phase composition

Fe_2O_3 content, mole	T, $^\circ\text{C}$	Phases	Fe_2O_3 content, mole	T, $^\circ\text{C}$	Phases
0.1	1000	wollastonite, periclase, diopside, akermanite	0.3	1000	wollastonite, diopside, akermanite, periclase
	1100	wollastonite, akermanite, diopside		1100	wollastonite, diopside, akermanite, cristobalite, periclase
	1200	diopside, cristobalite, wollastonite, akermanite		1200	Fe-diopside, cristobalite, wollastonite
0.2	1000	wollastonite, diopside, periclase, akermanite	0.4	1000	wollastonite, hematite, periclase, diopside, akermanite
	1100	diopside, wollastonite, akermanite, cristobalite, periclase		1100	wollastonite, hematite, diopside
	1200	Fe-diopside, cristobalite, wollastonite		1200	Fe-diopside, cristobalite, hematite

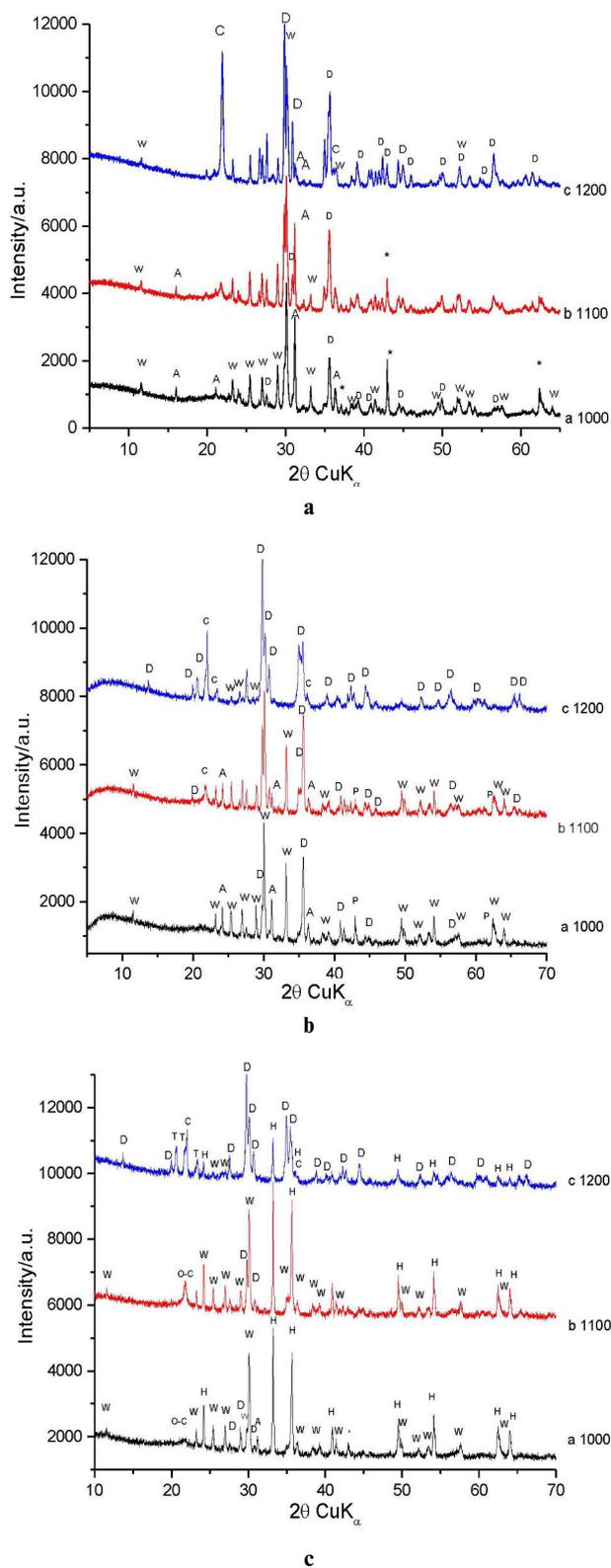


Fig. 1. Powder XRD patterns of samples $\text{CaO}\cdot 0.1\text{Fe}_2\text{O}_3\cdot 0.9\text{MgO}\cdot 2\text{SiO}_2$ (a), $\text{CaO}\cdot 0.3\text{Fe}_2\text{O}_3\cdot 0.7\text{MgO}\cdot 2\text{SiO}_2$ (b) and $\text{CaO}\cdot 0.5\text{Fe}_2\text{O}_3\cdot 0.5\text{MgO}\cdot 2\text{SiO}_2$ (c) sintered at 1000, 1100 and 1200°C: D – diopside, W – wollastonite, C – β -cristobalite, A – akermanite, P – periclase, H – hematite

(Table 1). X-ray diffraction patterns of the synthesized ceramic pigments are shown in Fig. 1.

Powder XRD patterns of the synthesized ceramic pigments $\text{CaO}\cdot x\text{Fe}_2\text{O}_3\cdot (1-x)\text{MgO}\cdot 2\text{SiO}_2$ at different temperatures (1000°C; 1100°C; 1200°C) and initial iron content $x=0.1, 0.2, 0.3, 0.4,$ and 0.5 mole revealed variations in phase composition depending on the initial iron oxide concentration and synthesis temperature.

At low temperature (1000°C), the predominate phase is wollastonite (CaSiO_3). Diopside ($\text{CaMgSi}_2\text{O}_6$) and akermanite ($\text{Ca}_2\text{MgSi}_2\text{O}_7$) are also formed in smaller quantities. Unreacted magnesium oxide remains in the form of periclase. At low content of iron oxide in the initial batch (0.1 mole to 0.3 mole), no separate iron oxide phases are detected, which means that iron oxide was successfully doped into the structure of the formed mineral phases. At higher content of iron oxide (≥ 0.4 mole), the excess of iron oxide leads to the formation of hematite ($\alpha\text{-Fe}_2\text{O}_3$).

With increasing the temperature to 1100°C, the amount of diopside increases and separate SiO_2 appears as β -cristobalite. Diopside and Fe-diopside predominates in samples sintered at 1200°C at iron concentrations above 0.1 mole. At the temperature of 1200°C, hematite is detected only at highest iron oxide content (0.5 mole). Probably at the concentrations of up to 0.3 mole, the whole amount of iron oxide is incorporated into diopside structure.

FT-IR analysis

FT-IR spectra of the iron-doped diopside ceramic pigments are shown in Fig. 2. Infrared peaks of diopside and other possible silicate phases peaks are summarized in Table 2.

Table 2
Infrared peaks of diopside and other possible silicate phases and their assignment

Diopside $\text{CaMgSi}_2\text{O}_6$	Wollastonite CaSiO_3	Akermanite $\text{Ca}_2\text{MgSi}_2\text{O}_7$	Type of vibration
1072	1090	1024	Si–O–Si stretching
960	1060	974	
920	1040	935	
860	1020	852	Si–O stretching
	967		
	904		
	679	675	Si–O–Si stretching
630	645	656	Si–O–Si bend
		586	Ca–O
515	470	486	O–Si–O bend
510	450		
470			O–Mg–O

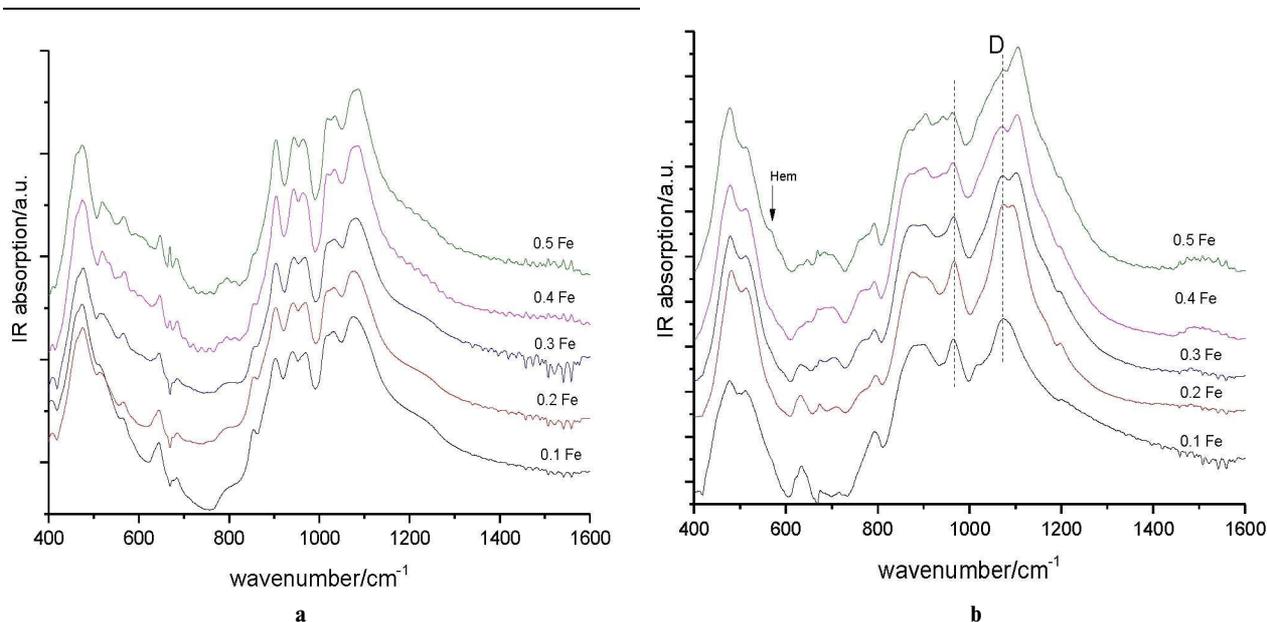


Fig. 2. FT-IR spectra of the iron-doped diopside ceramic pigments with different content of Fe_2O_3 synthesized at 1000°C (a) and 1200°C (b)

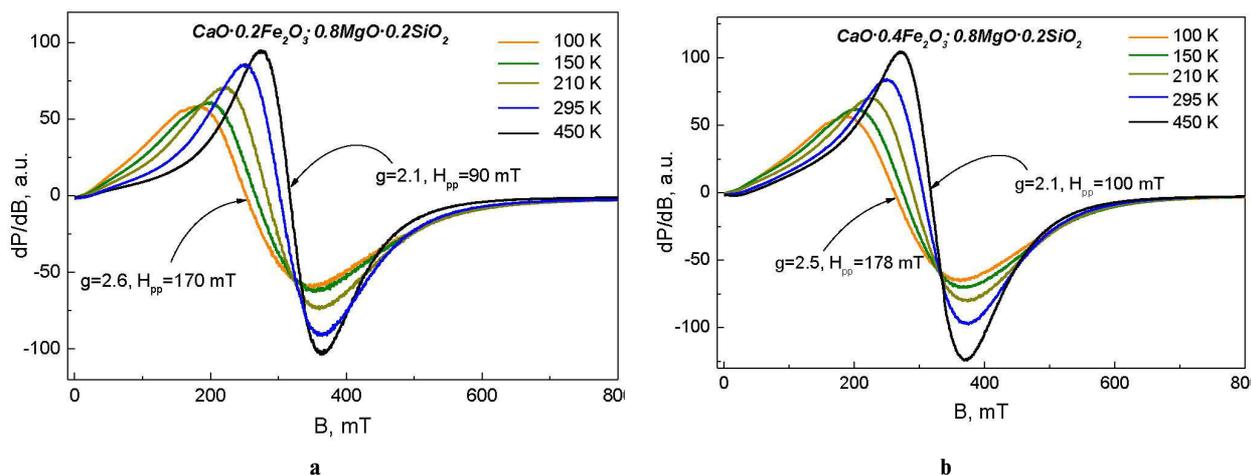


Fig. 3. EPR spectra of the following samples: $\text{CaO}\cdot 0.2\text{Fe}_2\text{O}_3\cdot 0.8\text{MgO}\cdot 0.2\text{SiO}_2$ (a) and $\text{CaO}\cdot 0.4\text{Fe}_2\text{O}_3\cdot 0.6\text{MgO}\cdot 0.2\text{SiO}_2$ (b) in temperature range of 100–450 K

Intense peaks at 1072, near 960 and 860 cm^{-1} are due to Si–O stretching in diopside. A peak at 515 cm^{-1} is characteristic for O–Mg–O bending vibrations in diopside. Similar peaks in akermanite should be at 1024, 974, 935 and 852 cm^{-1} due to Si–O–Si and Si–O stretching modes, while these peaks should be at 1060, 1000 and 900 cm^{-1} in wollastonite [14]. Peaks of these phases overlap and only the intensity ratio could be an indication for the composition change. The increase of the peak near 1200 cm^{-1} confirms an increase in the free SiO_2 content in the ceramic pigments obtained at 1200°C .

Electron paramagnetic resonance

EPR spectra of samples with the following composition $\text{CaO}\cdot x\text{Fe}_2\text{O}_3\cdot (1-x)\text{MgO}\cdot 0.2\text{SiO}_2$ ($x=0.2$

and 0.4 mole) are presented as the first derivative of absorption curve (dP/dB) versus magnetic field (B , mT) in Fig. 3.

The shape of considered signals and temperature dependence of their EPR parameters are an indication of the presence of effective exchange interactions occurring between Fe^{3+} ions. The recorded signals can be associated with the presence of a phase of Fe_2O_3 in the superparamagnetic state, i.e. with the existence of nanosized particles representing magnetic domains. The observed linear dependence (Fig. 4) is a proof that the nanodomains in the powder sample are homogeneous in size and composition.

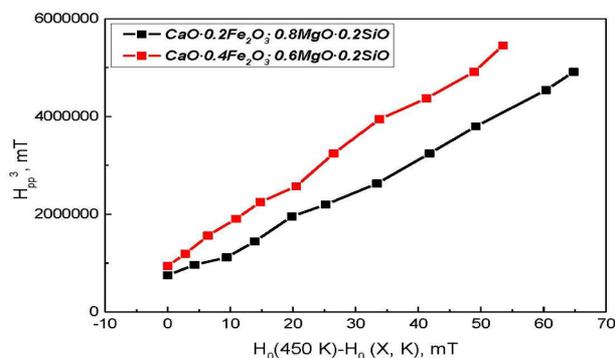


Fig. 4. Dependence of the third degree of peak to peak line width (ΔH_{pp}^3) on signal shift to weaker magnetic field with lowering the registration temperature for the samples $x\text{Fe}_2\text{O}_3 \cdot (1-x) \cdot \text{MgO} \cdot 2\text{SiO}_2$ ($x=0.2$ and 0.4 mole): B_0 – center field at 450 K, B_0 (K) – center field at given temperature in the temperature range of 100–450 K

Mössbauer spectroscopy

Mössbauer spectroscopy was used to identify the structural position of iron ions in the samples sintered at 1200°C with the Fe_2O_3 content of 0.2 mole and 0.4 mole. As can be seen from XRD phase composition (Table 1), the predominant crystalline

phase is Fe-diopside. The other accompanying phase is hematite, which is formed by an excess of iron above 0.3 mole Fe_2O_3 in the initial batch.

The experimental Mössbauer spectra of the samples with initial Fe_2O_3 content of 0.2 mole and 0.4 mole sintered at 1200°C are shown in Fig. 5.

Figure 5 shows a combination of doublets for the sample with 0.2 mole Fe_2O_3 and doublets and sextet for the sample with 0.4 mole Fe_2O_3 . The mathematical processing was performed according to the type of spectra: a model with three doublets was used for sample 16c, and three doublets and one sextet was used for sample 18c. The results of the spectrum processing are given in Table 3.

The obtained ultrafine parameters of the doublets can be explained according to ref. [15] as follows:

- Db1 Fe^{3+} in tetrahedral positions (Si^{4+}) in the ferri-diopside phase;
- Db2 Fe^{3+} in octahedral positions (M1) in the ferri-diopside phase;
- Db3 Fe^{2+} at M1 and / or M2 positions in the ferri-diopside phase;
- The sextet component in sample 18c has typical parameters for hematite.

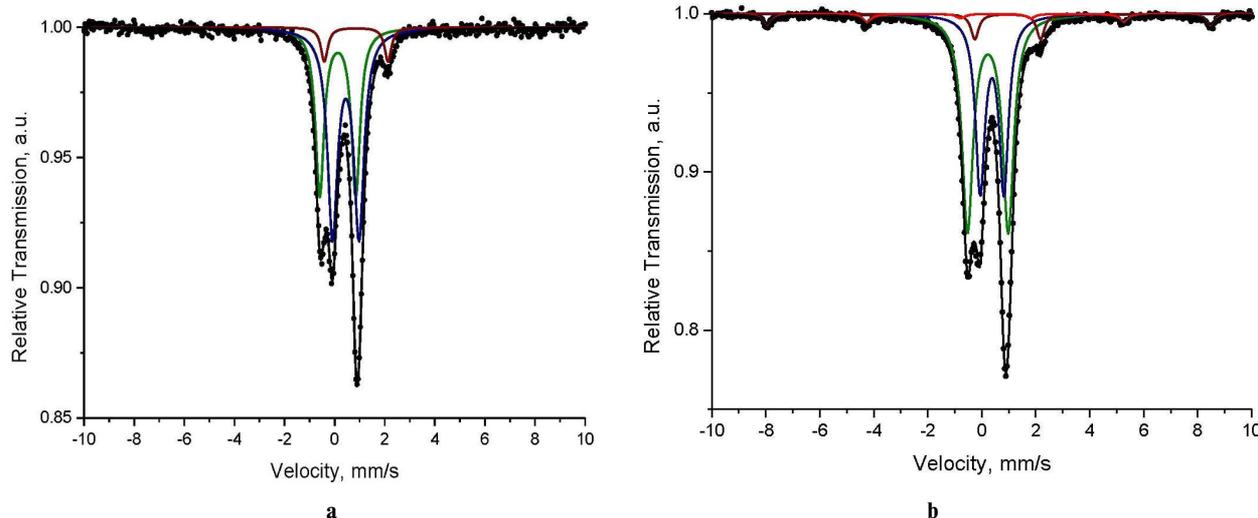


Fig. 5. Mössbauer spectra of the samples with the initial Fe content of 0.2 Fe_2O_3 (a) and 0.4 Fe_2O_3

Table 3

Mössbauer parameters of the studied samples

Sample	Components	δ , mm/s	$\Delta(2\varepsilon)$, mm/s	B_{hf} , T	Γ_{exp} , mm/s	G, %
16c, CaO·0.2Fe ₂ O ₃ ·0.8MgO·2SiO ₂ 1914-211	Db1-Fe ³⁺ _{tetra}	0.13	1.44	–	0.41	39
	Db2-Fe ³⁺ _{octa} , M1	0.45	0.06	–	0.49	55
	Db3-Fe ²⁺	0.86	2.54	–	0.33	6
18c, CaO·0.4Fe ₂ O ₃ ·0.6MgO·2SiO ₂ 1915-211	Db1-Fe ³⁺ _{tetra}	0.23	1.51	–	0.49	54
	Db2-Fe ³⁺ _{octa} , M1	0.31	0.88	–	0.42	37
	Db3-Fe ²⁺	0.96	2.44	–	0.34	4
	Sx-Fe ³⁺ _{octa} , α -Fe ₂ O ₃	0.39	–0.25	50.9	0.40	5

Table 4

Results obtained from the measurement of the color coordinates

Composition	Color	L*	a*	b*
CaO·0.2Fe ₂ O ₃ ·0.8MgO·2SiO ₂ (1000 ⁰ C)		50.8	22.5	27.3
CaO·0.2Fe ₂ O ₃ ·0.8MgO·2SiO ₂ (1100 ⁰ C)		59.8	19.9	29.9
CaO·0.2Fe ₂ O ₃ ·0.8MgO·2SiO ₂ (1200 ⁰ C)		64.9	6.0	31.1
CaO·0.3Fe ₂ O ₃ ·0.7MgO·2SiO ₂ (1000 ⁰ C)		44.5	24.1	25.8
CaO·0.3Fe ₂ O ₃ ·0.7MgO·2SiO ₂ (1100 ⁰ C)		49.6	23.7	26.3
CaO·0.3Fe ₂ O ₃ ·0.7MgO·2SiO ₂ (1200 ⁰ C)		62.1	6.1	32.6
CaO·0.4Fe ₂ O ₃ ·0.6MgO·2SiO ₂ (1000 ⁰ C)		41.5	25.7	23.6
CaO·0.4Fe ₂ O ₃ ·0.6MgO·2SiO ₂ (1100 ⁰ C)		43.9	24.2	24.6
CaO·0.4Fe ₂ O ₃ ·0.6MgO·2SiO ₂ (1200 ⁰ C)		58.6	7.4	31.9

Color measurement

Color is one of the most important indicators of the pigment quality. The expected color derived from Fe³⁺ ions substitution in the octahedral position in diopside is brown. The colors of the resulting ceramic pigments were defined according to universal CIELab system. In the CIELab system, the color coordinates are as follows:

L* (brightness), from absolute white L*=100 to absolute black L*=0,

a* – green color (-)/red color (+),

b* – blue color (-)/yellow color (+).

Table 4 shows the results of measuring the color indicators of the developed diopside pigments.

The results of the measurement of the color coordinates show that as the firing temperature increases from 1000°C to 1200°C, the color of the pigments changes from brown to beige. As the firing temperature increases, there is a tendency for the brightness L* to increase from 41.5–50.8 to 58.6–64.9%, while the coordinates a* and b* decrease. The amount of red color (a*) is greatest in the ceramic pigments sintered at lower temperature (1000°C and 1100°C), as it is seen from Table 4. The value of the color coordinate a* in this case is in the range from

+19.9 to +25.7. This can be explained by a significant increase in the SiO₂ phase content in the ceramic pigments obtained at 1200°C relative to the amount of Fe-diopside.

Conclusions

The brown and beige ceramic pigments were synthesized on the basis of stoichiometric diopside by the method of solid phase sintering. It was found that polyphase ceramic pigments are formed and the predominant phase is Fe-diopside. It was determined from the Mössbauer spectra that iron ions occupy different structural positions in the obtained Fe-diopside. It was also found that the color of the ceramic pigments depends on the quantitative ratio between the mineral phases obtained. At initial Fe₂O₃ content above 0.3 mole, the hematite is separated in the resulting ceramic pigments. The amount of free crystalline SiO₂ increases after firing at 1200°C, which reduces the intensity of the pigment color.

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СИНТЕЗ ТА ДОСЛІДЖЕННЯ ФЕРУМДІОПСИДНИХ КЕРАМІЧНИХ ПІГМЕНТІВ

Ц.І. Дімітров, Р.Х. Тіторенкова, О.В. Зайчук, Я.К. Цвєтанова

У статті досліджено мінералогічний склад, структурні особливості та зміну колірних параметрів залізовмісних пігментів діопсиду. Джерелом SiO₂ у таких пігментах був SiO₂·nH₂O із розміром частинок у діапазоні 2–7 мкм, що набагато активніші, ніж звичайний кварцовий пісок. Серія керамічних пігментів на основі стехіометричного діопсиду в системі CaO–Fe₂O₃–MgO–SiO₂ була синтезована твердо-тілним високотемпературним спіканням при 1000, 1100 та 1200°C. Одержані керамічні пігменти досліджували методом порошкової рентгенівської дифракції, інфрачервоної спектроскопії, електронної мікроскопії, електронного парамагнітного резонансу, мессбауєрівської спектроскопії. Кольорові характеристики вимірювали спектрофотометрично. Встановлено, що в умовах синтезу були одержані багатофазні керамічні пігменти, які містять у різній пропорції діопсид, волластоніт, кристобаліт, акерманіт і периклаз. Синтезовані керамічні пігменти мають коричневий і бежевий колір залежно від вмісту добавки Fe₂O₃, температури випалу та фазового складу. Розроблені пігменти використовують для забарвлення керамічної глазурі.

Ключові слова: керамічні пігменти, діопсид, мінеральний склад, колориметричні показники, синтез.

SYNTHESIS AND STUDY OF FERRI-DIOPSIDE CERAMIC PIGMENTS

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The article reports the mineralogical composition, structural features and changes in color parameters of iron-containing diopside pigments. The source of SiO₂ in these pigments was SiO₂·nH₂O with a particle size in the range of 2–7 μm, which is much more reactive than conventionally used quartz sand. A series of ceramic pigments on the basis of stoichiometric diopside in the system CaO–Fe₂O₃–MgO–SiO₂ was synthesized via solid-state high temperature sintering at 1000, 1100 and 1200°C. The resulting ceramic pigments were examined by powder X-ray diffraction, infrared spectroscopy, electron microscopy, electron paramagnetic resonance and Mössbauer spectroscopy. The color characteristics were measured using spectrophotometry method. It was found that multiphase ceramic pigments were obtained under the synthesis conditions, which contain diopside, wollastonite, cristobalite, akermanite and periclase in various proportions. The synthesized ceramic pigments are brown and beige in color, depending on the content of Fe₂O₃ additive, firing temperature and phase composition. The synthesized pigments can be used for staining of ceramic glazes.

Keywords: ceramic pigments; diopside; mineral composition; colorimetric indices; synthesis.

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