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Original Research

Sci Vis Vol 14 Issue No 3 July-September 2014 ISSN (print) 0975-6175 ISSN (online) 2229-6026

# Spectroscopic studies of Ho<sup>3+</sup> doped SiO<sub>2</sub>-TiO<sub>2</sub> nanoparticle for photonic applications

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Received 26 April 2014 | Revised 20 May 2014 | Accepted 30 May 2014

### ABSTRACT

The spectroscopic properties of Ho<sup>3+</sup> in SiO<sub>2</sub>-TiO<sub>2</sub> were investigated using optical absorption and fluorescence studies. The intensities of the electric dipole transitions of Ho<sup>3+</sup> in the SiO<sub>2</sub>-TiO<sub>2</sub> doped glasses were calculated from the absorption spectra in the visible region. From the experimental values of the oscillator strengths and calculated matrix elements, the Judd-Ofelt intensity parameters were calculated by least square analysis. The fluorescence spectral distribution of Ho<sup>3+</sup> has been recorded using 370nm excitation source. The emission bands observed near blue and green regions were assigned as  ${}^{5}G_{5}$ ,  ${}^{5}G_{6} \rightarrow {}^{5}I_{8}$ ,  ${}^{5}F_{3} \rightarrow {}^{5}I_{8}$  and  ${}^{5}F_{4}$ ,  ${}^{5}S_{2} \rightarrow {}^{5}I_{8}$  radiative transitions.

Key words: Fluorescence; Judd-Ofelt; oscillator strength; photoluminescence; sol-gel.

#### INTRODUCTION

Rare earth (RE) doped glasses have gained much interest owing to their attractive optical and physical properties. They have many potential and practical applications in the field of photonics, including solid state lasers and optical amplifiers in fiber optic communications.<sup>14</sup> The optical properties of RE ions arise mainly from the transitions between the different 4f<sup>n</sup> energy states, where n denotes the number of electrons on the 4f subshell.<sup>5</sup> The RE ion in the ground state may be excited to a higher energy state if it absorbs radiation having energy equivalent to the energy difference between the ground state and that particular excited state. Moreover, there is a possibility that the ion in the excited state may relax back to the lower energy levels thereby emitting radiation with energy equivalent to the energy difference between the two levels. This is known as photoluminescence (PL). Due to its favorable energy level structure Ho<sup>3+</sup> is one of the most attractive candidates for dopant among the RE ions. It shows strong luminescence at various wavelengths including the visible region.<sup>6</sup>

The structure and composition of the host glass is also an important factor as the optical properties of the RE dopant are influenced by

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their local environment.<sup>7</sup> The properties of the host glass such as density, refractive index, transparency, phonon-energy etc. have immense contribution to the luminescence properties. Silica, though it has high phonon energy, is an attractive host material as a result of its high chemical durability, thermal stability, and wide range of good transparency.8 Many studies have also shown that the presence of semi-conductor nanoparticles enhance the luminescence properties.<sup>9,10</sup> In the recent years studies on synthesis, optical properties and applications of rare earth (RE) doped nanomaterials have been attracting much interest. Among these luminescent materials, up conversion (UC) luminescent materials can convert near IR radiation into visible radiation by emitting a higher energy photon after absorbing multiple lower energy photons. In addition, the availability of low-cost near IR diode (980nm) has also stimulated research and applications in the area of UC luminescent materials. Materials doped with Er<sup>3+</sup> ions posses a favorable metastable energy level with longer lifetime excited states and  $Yb^{3+}$  ion have a large absorption cross-section around 980nm and can efficiently transfer the excitation energy to Er<sup>3+</sup> ions. TiO<sub>2</sub> have attracted much attention in recent years due to its interesting properties and many potential applications such as solar cells, sensors, photo catalysis etc.<sup>11, 12</sup> Many studies have been done on the photonic applications of TiO<sub>2</sub> as waveguides and as thin films.<sup>13, 14</sup>

In this paper, we report the synthesis and optical properties of  $\text{Ho}^{3+}$  and  $\text{TiO}_2$  nanoparticles co-doped in silica glass. The glass samples are prepared by sol-gel technique, which is a low temperature method of synthesis from a liquid phase and allow great control over the concentration of the dopants. The dependence of the luminescence property on the concentration of the TiO<sub>2</sub> is also shown. The optical studies are done by the use of Judd-Ofelt analysis.<sup>15-17</sup>

# **EXPERIMENTAL**

The glass sample was prepared by sol-gel technique where tetraethyl orthosilicate (TEOS)

is used as the glass precursor. Methanol was used as the solvent, nitric acid as the catalyst, holmium chloride and titanium isopropoxide (TIPO) as the sources for the  $Ho^{3+}$  and  $TiO_2$ dopants respectively. The dopants were mixed with methanol, nitric acid and water and stirred for 30 minutes. To this solution TEOS was added such that the ratio of TEOS, methanol, nitric acid and water is 16:70:4:10. The solution was further stirred for 1 hour till gelation started. Then the gel was put in a plastic caps and allowed to dry. After few days the gel shrunk to about have its size and solidified. Further drying and densification was done by annealing at 500° C.The glass sample obtained has thickness 0.2 cm and refractive index of 1.779.

The absorption spectrum of the prepared glass sample was recorded using Analytik Jena (model Specord 200) spectrophotometer and the photoluminescence (PL) spectra were recorded using iHR320 imaging spectrometer with the excitation wavelength of 370 nm.

#### **RESULTS AND DISCUSSIONS**

The absorption spectrum of the prepared sample is given in Figure 1. Absorption peaks were observed at wavelengths of 643, 538, 485, 471, 451 and 417 nm corresponding to the absorption from the ground state  ${}^{5}I_{8}$  to the excited states  ${}^{5}F_{5}$ ,  ${}^{5}S_{2}+{}^{5}F_{4}$ ,  ${}^{5}F_{3}$ ,  ${}^{5}F_{2}+{}^{3}K_{8}$ ,  ${}^{5}G_{6}$  and  ${}^{5}G_{5}$  respectively. The transitions within the *4f*<sup>e</sup> configuration of rare earth ions were analyzed using the Judd-Ofelt analysis.

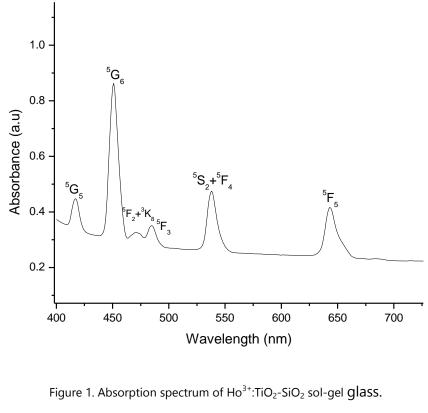
According to Judd-Ofelt Theory, the calculated oscillator strength of the electric dipole transition between two states  $|1^{N}SLJ \rightarrow |1^{N}SL'J' > can be expressed as^{10}$ 

$$f_{cal}^{ed} = \frac{8\pi^2 m c \overline{\nu} (n^2 + 2)^2}{3h(2J+1)9n} \sum_{\lambda=2,4,6} \Omega_{\lambda} |\langle l^N S L J || U^{(\lambda)} || l^N S' L' J' \rangle|^2$$
(1)

where m is the mass of electron, c is the ve-

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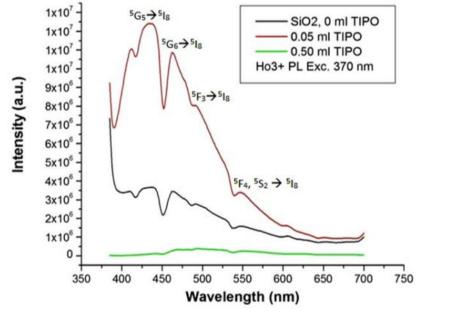


Figure 2. Photoluminescence spectra of Ho<sup>3+</sup>:TiO<sub>2</sub>-SiO<sub>2</sub> sol-gel glass with different concentrations of TiO<sub>2</sub>.

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locity of light, *h* is Planck's constant, *n* is the refractive index, *J* is the total angular momentum of the initial state,  $||U^{(\lambda)}||$  are the reduced matrix elements evaluated in the intermediate coupling approximation for transition  $|1^{N}SLJ \rightarrow ||1^{N}SLJ' \rightarrow | \alpha \alpha_{\lambda}$  are the Judd-Ofelt intensity parameters. The reduced matrix elements  $||U^{\lambda}||$  are known to be relatively host independent, so the values obtained by Carnall *et al.* are used in the calculations.<sup>18</sup>

From the absorption spectrum, the experimental oscillator strengths for transitions from the ground state to the excited states are determined using the relation

$$f_{exp} = 4.318 \times 10^{-9} \int \varepsilon(\overline{\nu}) d\nu \tag{2}$$

Where  $\varepsilon(\overline{v})$  is the molar absorptivity at energy  $\overline{v}$  cm<sup>-1</sup>. The experimentally determined values of oscillator strengths obtained from (2) are co-related with the theoretical expression given in equation (1) and the Judd Ofelt intensity parameters are evaluated using least square fit analysis. The Judd-Ofelt intensity parameters along with the experimental and calculated oscillator strengths are given in Table 1.

The JO intensity parameters give an insight into the local structure and bonding in vicinity rare earth ions.<sup>19</sup> The environment sensitive parameter  $\Omega_2$  indicates an amount of covalent bonding and the vibronic dependent parameter  $\Omega_6$  is related to the rigidity of the material. The ratio  $\Omega_4/\Omega_6$ - determine the spectroscopic quality of the material. In our sample we have  $\Omega_2 > \Omega_4 > \Omega_6$  and  $\Omega_4 / \Omega_6 = 1.197$ .

Once the three intensity parameters are known, other useful parameters such as spontaneous transition probability, branching ratio, radiative lifetime etc. can be calculated. The spontaneous electric dipole transition probability between the states  $|1^{N}SLJ\rangle$  and  $|1^{N}SLJ\rangle$  is given as

$$A_{ed}(SLJ, S'L'J') = \frac{16\pi^{3}e^{2}}{3h\varepsilon_{0}\overline{\lambda}^{3}(2J+1)} \frac{n(n^{2}+2)^{2}}{9}$$
$$\sum_{\lambda=2,4,6} \Omega_{\lambda} |\langle l^{N}SLJ || U^{(\lambda)} || l^{N}S'L'J' \rangle|^{2}$$
(3)

where  $\lambda$  is the average wavelength of transition.

The fluorescence branching ratio can be calculated from

$$\beta(SLJ, S'L'J') = \frac{A(SLJ, S'L'J')}{\sum_{S'L'J'} A(SLJ, S'L'J')}$$
(4)

where the summation in the denominator is for all the transitions from the excited state  $|1^{N}SLJ\rangle$  to the lower energy states.

The radiative lifetime of an excited state  $|1^{N}SLJ\rangle$  is given as

$$\tau(SLJ) = \left(\sum_{S'L'J'} A(SLJ, S'L'J')\right)^{-1}$$
(5)

Using these equations, the transition probability, branching ratio and radiative lifetime of the various states are calculated and presented in

Transition	Energy (cm <sup>-1</sup> )	$f_{exp}$ (×10 <sup>-6</sup> )	f <sub>cal</sub> (×10 <sup>-6</sup> )		
${}^{5}I_{8} \rightarrow {}^{5}G_{5}$	23981	1.793	1.999		
${}^{5}I_{8} \rightarrow {}^{5}G_{6}$	22173	10.396	10.385		
${}^{5}I_{8} \rightarrow {}^{5}F_{2} + {}^{3}K_{8}$	21231	0.331	1.172		
${}^{5}I_{8} \rightarrow {}^{5}F_{3}$	20619	0.808	0.931		
${}^{5}I_{8} \rightarrow {}^{5}F_{4} + {}^{5}S_{2}$	18587	3.225	2.958		
${}^{5}I_{8} \rightarrow {}^{5}F_{5}$	15552	2.464	2.186		
$\Omega_2 = 1.972 \times 10^{-20} \text{ cm}^2$ , $\Omega_4 = 1.469 \times 10^{-20} \text{ cm}^2$ , $\Omega_6 = 1.227 \times 10^{-20} \text{ cm}^2$					

Table 1. Oscillator strengths and Judd-Ofelt intensity parameters for Ho<sup>3+</sup>:TiO<sub>2</sub>-SiO<sub>2</sub> sol-gel glass.

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Tran	sition	Average Energy (cm <sup>-1</sup> )	A <sub>ed</sub> (s <sup>-1</sup> )	β	Τ (μs)
${}^{5}G_{5} \rightarrow {}^{5}I_{6} {}^{5}I_{5} {}^{5}I_{4} {}^{5}I_{4} {}^{5}F_{5} {}^{5}I_{4} {}^{5}F_{5} {}^{5}I_{5} {}^{5}I_{4} {}^{5}F_{5} {}^{5}I_{5} {}$	<sup>5</sup> l <sub>8</sub>	23041	3265	0.482	
	<sup>5</sup> I <sub>7</sub>	17986	2627	0.388	
	<sup>5</sup> I <sub>6</sub>	14472	630	0.093	147.60
	<sup>5</sup> I <sub>5</sub>	11905	100	0.015	
	<sup>5</sup> I <sub>4</sub>	9839	21	0.003	
	<sup>5</sup> F <sub>5</sub>	7594	132	0.019	
${}^{5}I_{8}$ ${}^{5}I_{7}$ ${}^{5}I_{6}$ ${}^{5}I_{5}$ ${}^{5}I_{4}$ ${}^{5}F_{5}$	<sup>5</sup> l <sub>8</sub>	21692	12883	0.860	66.76
	<sup>5</sup> I <sub>7</sub>	16636	1682	0.112	
	<sup>5</sup> I <sub>6</sub>	13122	181	0.012	
		10555	33	0.002	
	<sup>5</sup> I4	8489	1	0.000	
		6244	200	0.014	
$F_{3} \rightarrow \frac{5_{18}}{5_{17}}$	<sup>5</sup> l <sub>8</sub>	20408	3321.71	0.816	257.07
	<sup>5</sup> I <sub>7</sub>	15353	339.96	0.084	
	<sup>5</sup> I <sub>6</sub>	11839	246.78	0.061	
	<sup>5</sup> I <sub>5</sub>	9272	132.12	0.032	
	<sup>5</sup> l <sub>4</sub>	7206	22.53	0.006	
	<sup>5</sup> F <sub>5</sub>	4961	6.62	0.001	

Table 2. Calculated radiative parameters for Ho<sup>3+</sup>:TiO<sub>2</sub>-SiO<sub>2</sub> sol-gel glass.

Table 2 above.

From the  $\beta$  values in Table 2, we can see that the transitions from the excited levels to the ground state <sup>5</sup>I<sub>8</sub> are the dominant transitions. The photoluminescence (PL) spectra of Ho<sup>3+</sup>:TiO<sub>2</sub>-SiO<sub>2</sub> glass for different TiO<sub>2</sub> contents is given in Figure 2. The emission bands near 434nm, 461nm and 490nm (blue) are assigned as due to the  ${}^{5}G_{5} \rightarrow {}^{5}I_{8}$ ,  ${}^{5}G_{6} \rightarrow {}^{5}I_{8}$  and  ${}^{5}F_{3} \rightarrow {}^{5}I_{8}$  optical transitions respectively.<sup>20</sup> The emission bands at green and red comparatively weak compared to blue bands are also observed at 547nm and 620nm correspond to  ${}^{5}F_{4}$ ,  ${}^{5}S_{2} \rightarrow {}^{5}I_{8}$ and  ${}^{5}F_{5} \rightarrow {}^{5}I_{8}$  transitions and the radiative parameters are not calculated for these weak transitions. It is observed that the intensity of blue transition  ${}^{5}G_{5} \rightarrow {}^{5}I_{8}$  is the highest among all the transitions. The TiO<sub>2</sub> concentration is varied by varying the amount of TIPO added in the preparation process.

It is observed that the PL intensity initially increases as the TiO<sub>2</sub> concentration increases, reaches a maximum, and then decreases with further increase in concentration. One possible explanation is that the TiO<sub>2</sub> particles act as network modifiers and increase the concentration of Si dangling and oxygen vacancy in the glass network.<sup>21</sup> In this way, more electrons and holes can be easily excited and radiant recombinations are increased. Moreover, the interesting optical properties of the TiO<sub>2</sub> nanoparticles must be a significant factor. The presence of TiO<sub>2</sub> may assist in the excitation of the RE ions by energy transfer from the TiO<sub>2</sub> particles to the RE ions. The decrease in PL with further increase in concentration may be due to the clustering of  $TiO_2$ .

# CONCLUSION

In the conclusion, the absorption and fluorescence spectra for  $\text{Ho}^{3+}$  in  $\text{SiO}_2\text{-TiO}_2$  have been investigated. Estimate reveals that the optical parameters are better to that for  $\text{Ho}^{3+}$  in other glasses reported because of the improvement of the host glass. The ability to use semiconductor nanocrystals as a sensitizer doped with  $\text{Ho}^{3+}$ ions in  $\text{SiO}_2\text{-TiO}_2$  glass have been studied. Furthermore, the figures of merit associated with these  $\text{TiO}_2$ nanocrystals are of sufficient magnitude to quantify this nanocomposite as a high gain material.

#### ACKNOWLEDGEMENT

Authors are thankful to Department of Science and Technology (DST)for financial assistance (no. SR/S2/LOP-0039/2010). One of us, A. Fanai, is also thankful to DST for awarding research fellowship.

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